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Morphology of Oligomers and Their Network Polymers

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SUMWARY

The morphology of di(4-methacryloxyethylene carbamate toluylen-2-carbamic) esters of oligoethylene glycols (OUM) with different molecular weight, of their network polymers and of model oligomers without methacrylic groups was studied. The supermolecular structure of oligomers is found to depend on molecular weight. Globular structure is typical for low molecular weight oligomers with $M_n\neq 1200$. The structure of crystalline samples with $M_n = 1600-3600$ is lamellar. The latter is retained on melting. It is transformed into globular structure at a temperature much higher than the melting point. A correlation exists between the morphology of polymerizable oligomers and their network polymers.

The process of network formation from oligomers is extremely complicated. Irrespective of the mechanism it is influenced by MWD, FTD,intermolecular interaction and a number of other factors. The interconnection between the morphology of polymerizable oligomers **and** the supermolecular structure of their crosslinked polymers has not been sufficiently studied.Information about this connection facilitates phenomenological understanding of the network formation process.

EXPERIMENTAL

The OUM's and model compounds (OUE) were prepared by reaction of different molecular weight ollgoethylene glycols with 2-isocyanate toluylene-4-methacryloxyethylene (or 4-ethyl) urethane (Tabl.1) (ZEMSKOVA et al. 1973); network polymers were obtained by radical polymerization of OUM's (MATVEEVA et al. 1974). The supermolecular structure of liquid oligomers and melts was fixed by fast freezing of samples with liquid nitrogen; micrographs of replicas of oligomer and polymer samples were obtained with an EUMB-100B electron microscope; the fracturing of glassy oligomers and deposition of the carbon film on the fracture surface were carried out under vacuum at $-150 - -160$ °C (MATVEEVA,

KISELEV et al. 1978); platinum-carbon replicas of polymer samples were prepared by procedure (KISELEV et al. 1966); a small-angle X-ray scattering study was carried out with KRM-1 using CuK $_{\boldsymbol{\alpha}}$ -emission; the experimental data were corrected by a known method (GUINIER, FOURNET 1955).

TABLE 1

Characteristics of OUM's, model oligomers and mixtures of latters

RESULTS AND DISCUSSION

The oligomers depending on molecular weight are viscous liquids or waxy compounds. The supermolecular structure of glassy low molecular weight OUM ($\overline{\mathbb{M}}_n \neq 1200$) is characterized by the presence of globular formations ranging from 100 to 500A.For instance, the size of globules for OUM with $\texttt{M}_{\texttt{n}}$ = 1200 is 180-200A (Fig.1). The morphology of crystalline OUM with $M_{\Omega}=1600-3600$ is lamel<u>l</u>ar.The thickness of lamellae of the oligomer with M_n =1600 is 100-130A;of OUM with M_n =2250, 150- -180 A and of OUM with $M_{\rm B}$ =3600, to 80-100 A (Fig.1). The lamellar structure of OUM is retained on melting and on subsequent cooling down to the room temperature.In some cases the melts do not crystallize and remain in the liquid-crystalline state (Fig.2).The dependence of

Fig.1.Morphology of different molecular weight ${\tt oll}$ gomers ${\tt oll}$

oligomer morphology on the temperature was studied in order to estimate the stability of a supermolecular organization of oligomers and to find the correlation

 $Fig.2$.Morphology of oligomers with $M_{n} = 2250$ in a liquid- melting points are crystalline state.

between the morphology of oligomers and their network polymers.Model oligomers without methacrylic groups were synthetized to exclude the polymerization of OUM's. Micrographs of fracture replicas of vitrified oligomer melts heated to \sim 10°
and \sim 100° above the The thickness of la-

melae of vitrified melts is close to the values typical for the same oligomers in the crysballine state.With further increase of the temperature, globules emerge on the surface of the lamellae.The lamellar structure is completely transformed to a globular one at a temperature much higher than the melting point (MATVEEVA, KISELEV et al. 1979).

Fig.3. Temperature transformation of the morphology of the oligomer with $\overline{M}_n = 3400$.

Fig. 4. Small-angle X-ray scattering curves for oli- the e
gomers with $\overline{\text{M}}$ =1200 (2) data. gomers with $\overline{M}_{\overline{k}}$ 1200 (2)
and $\overline{M}_{\overline{n}} = 2250$ (1).

Small-angle X-ray scattering curves display a distinct maximum which indicates the presence of oligomer organization at the supermolecular level. The position of the maximum varies with the molecular weight of the oligomer (Fig.4).The value of the large period for the oligomer with $\overline{M}_{n} =$ 1200 is 150A and for
the oligomer with $\overline{M}_{n} =$ $= 2250,170$ Å.The results of the small-angle X-ray scattering study of oligomer morphology confirm

The supermolecular organization of the mixtures

of oligomers is controlled by the molecular weight and the ratio of initial components. The presence of lamellar structure (thickness \sim 100A) decorated by globules of the low molecular oligomer (d\1OOA) <u>i</u>s characteristic of the mixture of the OUE's with M_{n=}600 and 3400 enriched by the high molecular component (weight ratio 1.20) (Fig.5a).The morphological pattern essentially varies when a part of the low molecular weight oligomer increases. Separate lamellar incorporations typical of the high molecular ingradient are observed for the mixture of components taken in equal parts by weight. At the same time the background of the sample consists of globules (homogeneous in size) of the low molecular weight oligomer (Fig.5b).

Fig.5.The morphology of mixtures of oligomers as a function of molecular weights and composition.

The increase in molecular length of the low molecular part of the mixture (\overline{M}_p =1040 and 3400) practically do not influence the morphology of the high molecular weight component at a small content of the former (weight ratio 1:20). In this case the distinctly formed lamellar structure is retained; the thickness of lamellae is close to the value for the crystalline state of this component; however,accumulation of the globules and erosion of the lamella edges in individual separate regions are observed (Fig.5c).The variation in the oligomer ratio in the mixture towards the low molecular weight component (weight ratio 1:1)leads to essential looseness of the structure - the thickness of the lamellae increases and the sharpness of the lamella edges disappears (Fig.5d).

Thus, the morphology of the oligomers is controlled by their molecular welght. The studied oligomers are characterized by the chain regularity, by the terminal groups capable to form H-bonds and by the narrow MWD. All that ensure preservation of the order in the oligomer melts at temperatures higher than the melting points. Fractionation in size of the macromolecules is observed for mixtures of different molecular weight oligomers prepared by mixing of the solution of the components and by subsequent evaporation of the solvent.As a result the morphological pattern represents a combination of the supermolecular elements characteristic of the initial oligomers.

There exists a connection between the supermolecular structure of crosslinked polymers and the morphology of initial oligomers. Network polymers prepared by curing low molecular weight oligomers are composed of globules and the different formations consisting of coalesced globules. The morphological pattern of polymers obtained from oligomers having a lamellar structure does not differ from the structure of the initial oligomers. The lamellar structure is observed also for an amorphous polymer with the value of interstitual chains $\overline{M}_{c} = 1600$ (Fig.6).

The supermolecular structure network polymer obtained by curlng the mixture of oligomers with different molecular weight is characterized by the presence of various structural elements. Regions of globular and "spherulite"-like formations have been found in an amorphous polymer based on the mixture of oligomers $_{\rm M}$ =670 and $_{\rm M}$ =6600 taken in equal parts by weight (Fig.7~,b). A polymer obtained by curing the mlxture of equal parts by weight of OUM's with M $_{\odot}$ = 750.1600 $\,$ and 3600 is amorphous and has a globular"- la structure with eroded boundaries after two-week standing at room temperature. After more than a three-year standing this polymer crystallizes to a small degree of crystallinity (<10%) and its structure remains

Fig.6.Correlation between morphology of oligomers and their network polymers.

globular, but the lamellae become more dictinct (Fig. 7c). A similar phenomenon is observed for the polymer prepared from a mixture consisting of eight oligomers with narrow MWD and different molecular weight ranging from $M_h = 670$ to $M_n = 6600$. After two-week standing this polymer is~morphous and has a globular structure. After long standing it partly crystallizes (~ 10%); rather well formed lamellae are present in its structure together with globular formations.

Fig.7.Morphology of network polymers based on mixtures of oligomers.

The following phenomenological pattern of the formation of crosslinked polymers by polymerization of \propto,ω bis unsaturated oligomers may be suggested (BERLIN 1977: MATVEEVA 1977):

During crosslinking polymerization of low molecular weight oligomers the resulting primary polymethacrylic chains, having an oligomeric pendant in every unit, coil and aggregate with each other. The growth of macrochains continues due to consumption of the oligomer immobilized in the coils and of the methacrylic groups in the pendant oligomer blocks. The macrochains branch out, form cycles, and crosslink together. As a result, stabilization of these aggregates takes place. In the course of thia process the density of aggregate particles gradually increases as a result of their contraction. Simultaneously the syneresis of the soluble polymerizate may take place.The individual particles become linked together by polymethacrylic tie-chains and oligomeric blocks. Nhen the number of such particles and of the branched and tie-chains becomes sufficient for immobilizing the liquid reactant, there occurs gelation, and the system loses its fluidity. As a result, a microheterogeneous three-dimensional crosslinked system is formed and highly crosslinked microparticles are interconnected by zones having an essentially lower crosslinking density.

Fig.8. Scheme of the network formation from oligo $mers.$

In the case of crystalline oligomers their transition on melting to the liquid-crystalline state with lamellar structure is observed. In such systems the polymerization process develops in the interlayer zones where terminal functional groups are concentrated. The mobility of the terminal groups of oligomeric molecules in these zones is quite sufficient for realization of the process of propagation and conformational transformations of growing primary macrochains. As a result of three-dimensional polymerization, the terminal groups of oligomeric molecules become crosslinked and lamellar structure of the liquid is fixed in the polymer.

The process of the network formation in the case of oligomeric mixtures includes elements of the mechanisms described above. The supermolecular structure of such polymers is either completely globular, or it consists of combinations of globular formations with lamellae not well defined or of other structural elements. However, in spite of considerable disorder and the limited mobility of interstitual chains, a gradual increase of order in the system, a crystallization of long-chain oligomeric blocks, and a formation of lamellar structure regions may be observed in the network polymer.

REFERENCES

BERLIN **A.A.,** The 1-st All-Union Conf. on Chemistry and Physico-Chemistry of Polymerizable Oligomers, Chernogolovka, 1977, P.3 (in Russian). GUINIER **A., FOURNET G.** Small-Angle X-Ray Scattering, N.Y., 1955. KISELEV M.R., EVKO E.I., LUKJANOVICH V.M., Zavodskaja Lab., No .2, 201 (1966). MATVÉEVA N.G., ZEMSKOVA Z.G., SIVERGIN YU.M., BERLIN A.A., Vysokomol.soedin., ser.A <u>16</u>, 588 (1974). MATVEEVA N.G., The 1-st All-Union Conf. on Chemistry and Physico-Chemistry of Polymerizable Oligomers, Chernogolovka, 1977, p.277 (in Russian). MATVEEVA N.G., KISELEV M.R., PLAVNIK M.G., KUSHNEREV M.Ya., BERLIN A.A. and ZUBOV P.I., Vysokomol.soedin., Ser. A₂₀, 1080 (1978). MATVEEVA'N.G., KISELEV M.R., ZUBOV P.I., BERLIN A.A. The 2-nd All-Union Conf. on Chemistry and Physico- -Chemistry of Oligomers, Chernogolovka, 1979, P.7 (in Russian). ZEMSKOVA Z.G., MATVEEVA N.G. and BERLIN A.A. Vysokomol. soedin., Ser. A 15, 724 (1973).

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